



Short Communication

Multistep evolution from a metal–organic framework to ultrathin nanosheets

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Two-dimensional (2D) materials and ultrathin nanosheets have attracted tremendous research interests [1–4]. Exfoliation of porous coordination polymers (PCPs) or metal–organic frameworks (MOFs) with 2D coordination structures is emerging as a viable strategy for preparation of new types of 2D materials [5–8]. The relatively weak bonding within and strong interactions between 2D coordination networks are the main problems for successful exfoliation [9]. Pretreatments of the layered MOFs to expand the interlayer distances can be helpful [10–12]. Capturing the exfoliation intermediates and visualizing their structures are valuable for understanding the exfoliation mechanism and confirming the structures of the final 2D materials [11,12].

While crystal-to-crystal and single-crystal-to-single-crystal structural transformations of MOFs have been widely reported, those involving reconstitution of coordination bonds and topologies (e.g., from 3D to 2D) are rare, because the tension and shearing forces tend to degrade crystallinity, especially single-crystallinity [13–17]. Two examples reported single-crystal structures of the intermediate states of MOF exfoliation [11,12]. Jiang group and Zhou group [11] treated a MOF crystal with 2D coordination structure by 4,4'-dipyridyl disulfide to expand the interlayer distance, and then exfoliated the latter by trimethylphosphine. Single-crystal structure showed that the 4,4'-dipyridyl disulfide served as a terminal rather than bridging ligand. Aida group [12] showed that a MOF crystal with 2D coordination network can be exfoliated in different solvents to form ultrathin nanosheets with thickness of either 1.3 or 2.0 nm. Single-crystal structures of the intermediates showed that the disordered solvent molecules inserted between monolayers and bilayers, explaining the different thicknesses of

the products. Recently, we showed that pillared-layer MOFs may be good candidates for preparation of 2D materials, by virtue of selective breaking the pillar ligands [18]. Here, we report a new pillared-layer MOF which can be easily exfoliated to form ultrathin nanosheets with multiple single-crystalline or polycrystalline intermediate states.

Solvothermal reaction of $\text{Cd}(\text{NO}_3)_2 \cdot 4\text{H}_2\text{O}$ with 2,3-dihydroxy-1,4-benzenedicarboxylic acid (H_4dihbdc) in mixed solvent containing *N,N*-dimethylformamide (DMF) and H_2O afforded crystals of $(\text{H}_3\text{O})_2[\text{Cd}_4(\text{dihbdc})_2(\text{H}_2\text{dihbdc})(\text{DMF})_4] \cdot \text{H}_2\text{O}$ (**1**) crystallizing in the *C2/c* space group (Table S1 online), which has a 3D anionic pillared-layer structure consisting of neutral wavelike $\{\text{Cd}_2(\text{dihbdc})\}$ layers running across the *bc*-plane (Fig. 1) and anionic $\text{H}_2\text{dihbdc}^{2-}$ pillars (Fig. 2a). In the $\{\text{Cd}_2(\text{dihbdc})\}$ layer, each dihbdc^{4-} coordinates with 6 Cd^{2+} ions with 9 coordination bonds (3 pairs are bidentate chelating) and each Cd^{2+} ion coordinates with 4 or 5 oxygen atoms from three dihbdc^{4-} ligands (Fig. S1 online). The vacant coordination sites of Cd^{2+} are capped by one oxygen atom from DMF and one carboxylate oxygen atom of the $\text{H}_2\text{dihbdc}^{2-}$ pillars at opposite sides of the layer to furnish the octahedron or decahedron coordination geometry (Fig. S1 online). The 3D pillared-layer structure has an interlayer separation of 10.2 Å and a pillar inclining angle of 79.7° (Figs. 2a and S2 (online)). There are 2D intersecting channels between the layers, which are occupied by the $\text{H}_3\text{O}^+/\text{H}_2\text{O}$ guests. Thermogravimetry and powder X-ray diffraction (PXRD) showed that after removal of the guest and coordinated solvent molecules (Fig. S3 online), the host of **1** loses crystallinity (Fig. S4 online).

In water, **1** quickly transforms to a new structure $[\text{Cd}_4(\text{dihbdc})_2(\text{H}_2\text{O})_8] \cdot 2\text{H}_2\text{O} \cdot \text{H}_4\text{dihbdc}$ (**4**) as microcrystals (Fig. S5 online). Further, $(\text{H}_3\text{O})_2[\text{Cd}_4(\text{dihbdc})_2(\text{H}_2\text{dihbdc})(\text{H}_2\text{O})_4] \cdot \text{H}_2\text{O}$ (**2**) and $(\text{H}_3\text{O})[\text{Cd}_4(\text{dihbdc})_2(\text{H}_3\text{dihbdc})(\text{H}_2\text{O})_7] \cdot 2\text{H}_2\text{O}$ (**3**) were captured as single-crystalline intermediates between **1** and **4**, by using

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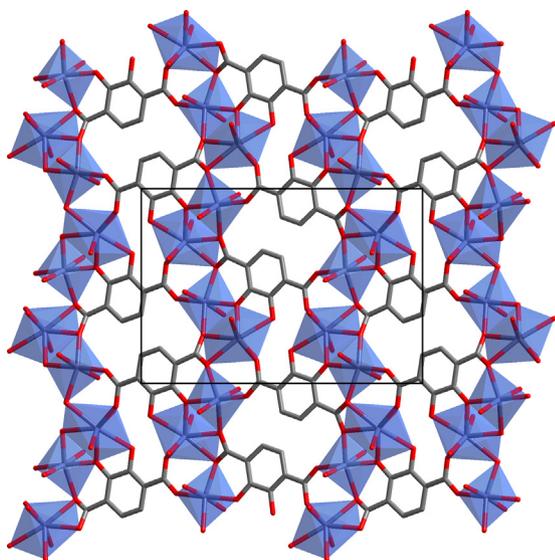


Fig. 1. (Color online) The $\{Cd_2(dhbdcc)\}$ layer in **1**. The coordinated oxygen atoms from DMF and $H_2dhbdcc^{2-}$ are added to show the coordination geometries of the metal ions as semi-transparent polyhedra.

milder reaction conditions, i.e., immersing **1** in methanol (containing ca. 0.05 wt% H_2O) and exposing **2** in H_2O vapor, respectively (Figs. S6–S8 online). **2** and **3** can also transform to **4** in water (Fig. S9 online), and **2–4** can all transform back to **1** in DMF vapor (Fig. S10 online).

2 and **3** both crystallize in the same space group as **1** (Table S1 online). The b - and c -axes of **2** and **3** change less than 3% from those of **1**, indicating that the $\{Cd_2(dhbdcc)\}$ layers are kept with little distortion (Figs. S11 and S12 online). In contrast, the a -axis significantly shrinks from 24.0 Å for **1** to 17.3 Å for **2** and further to 16.0 Å for **3**, corresponding to the reduction of the interlayer separation from 10.2 Å for **1** to 8.4 Å for **2** (Fig. 2b) and further to 7.6 Å for **3** (Fig. 2c). Also, β reduces from 122.1° for **1** to 105.0° for **2** and 108.7° for **3** (Table S1 online), indicating the change of inclining angle of the pillars and sliding between adjacent $\{Cd_2(dhbdcc)\}$ layers. Structural refinements of the single-crystal structures showed that the metal ions and organic ligands in **2** have the same coordination modes as for **1**, except that the coordinated DMF molecules are replaced by H_2O molecules (Fig. S1 online). **2** retains the 3D anionic pillared-layer structure, but the pillar inclining angle significantly reduces from 79.7° to 45.2° (Fig. S2 online), accounting for the reduction of interlayer distance, which can be explained by the very different molecular sizes of the coordinated DMF and H_2O molecules.

In **3**, the coordination modes of metal ions and $dhbdcc^{4-}$, as well as $\{Cd_2(dhbdcc)\}$ layers were retained (Figs. S1, S11 and S12 online). However, **3** turned into a 2D coordination structure (Fig. 2c). Each $H_2dhbdcc^{2-}$ pillar in **1** and **2** uses its two carboxylate groups to coordinate two pairs of Cd^{2+} ions from two layers, but in **3** each $H_3dhbdcc^-$ “pillar” uses only one carboxylate group in the monodentate mode to coordinate with one Cd^{2+} (Fig. S1 online). In other words, one bidentate anchoring end of $H_2dhbdcc^{2-}$ completely leaves the layer (replaced by two H_2O ligands) and simultaneous protonated by the guest H_3O^+ to form $H_3dhbdcc^-$, and another bidentate anchoring end becomes monodentate (replaced by one H_2O ligand). The dangling $H_3dhbdcc^-$ is almost parallel with the $\{Cd_2(dhbdcc)\}$ layer, with an inclining angle of 9.8° (Fig. S3 online).

Microcrystalline **4** displays weak and simple PXRD pattern, indicating partial loss of long-range order and preventing us to experimentally determine its crystal structure. According to the

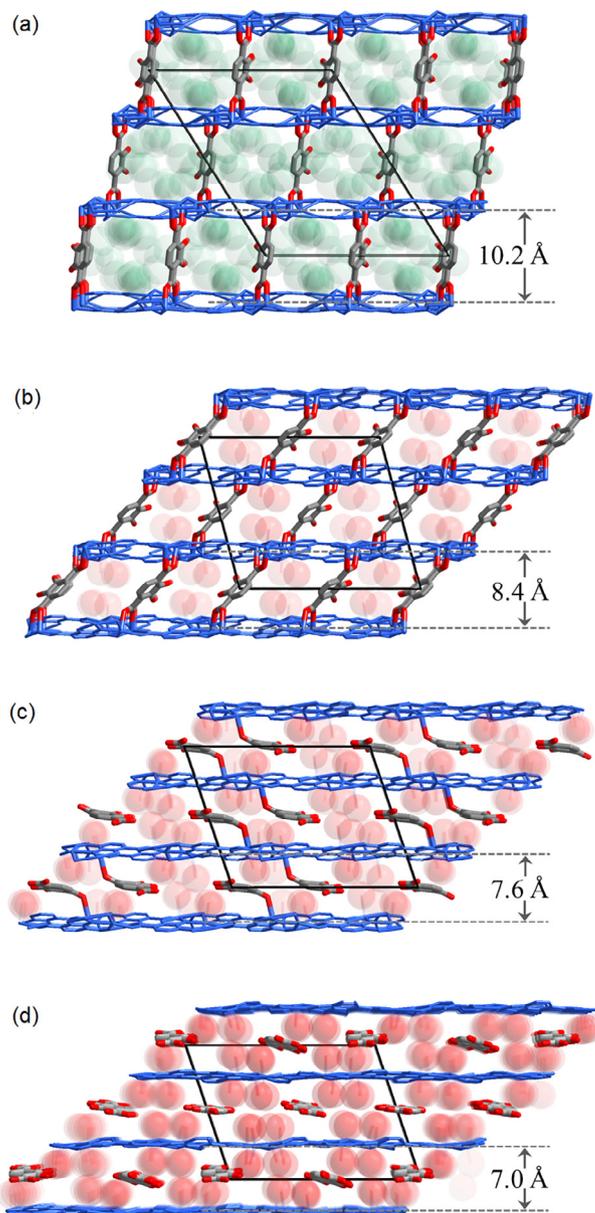


Fig. 2. (Color online) Crystal structures of (a) **1**, (b) **2**, (c) **3**, and (d) **4** viewing along the b -axis. For clarity, $\{Cd_2(dhbdcc)\}$ layers are drawn in blue, $H_2dhbdcc^{2-}/H_3dhbdcc^-/H_4dhbdcc$ pillars are drawn as thick sticks, and coordinated solvent molecules are drawn in the space-filling and semi-transparent mode. Hydrogen atoms, guest water and hydronium are omitted.

main diffraction peak of **4** at 12.7°, the interlayer distance can be calculated as 7.0 Å. Considering the situation of the “pillars” as seen in **3**, they may leave the $\{Cd_2(dhbdcc)\}$ layers and even the crystals. Infrared (IR) spectra showed that, **4** retains the C–O(H) stretch vibration and CO–H bending vibration (Fig. S13 online), meaning that the “pillars” are still in the crystals. We simulated the crystal structure of **4** (Fig. 2d), by replacing the “pillars” in the crystal structure of **3** with H_2O ligands, setting the interlayer separation to 7.0 Å, optimizing the structure with molecular mechanics, and finally load $H_4dhbdcc$ and H_2O guests into the structure by the Grand Canonical Monte Carlo method. If the interlayer separation was set shorter, $H_4dhbdcc$ guest molecules cannot be loaded. This further confirmed the presence of $H_4dhbdcc$ guest molecules between the layers in **4**, which could help separating the layers or weakening the interlayer interactions for preparation of ultrathin nanosheets [11,12].

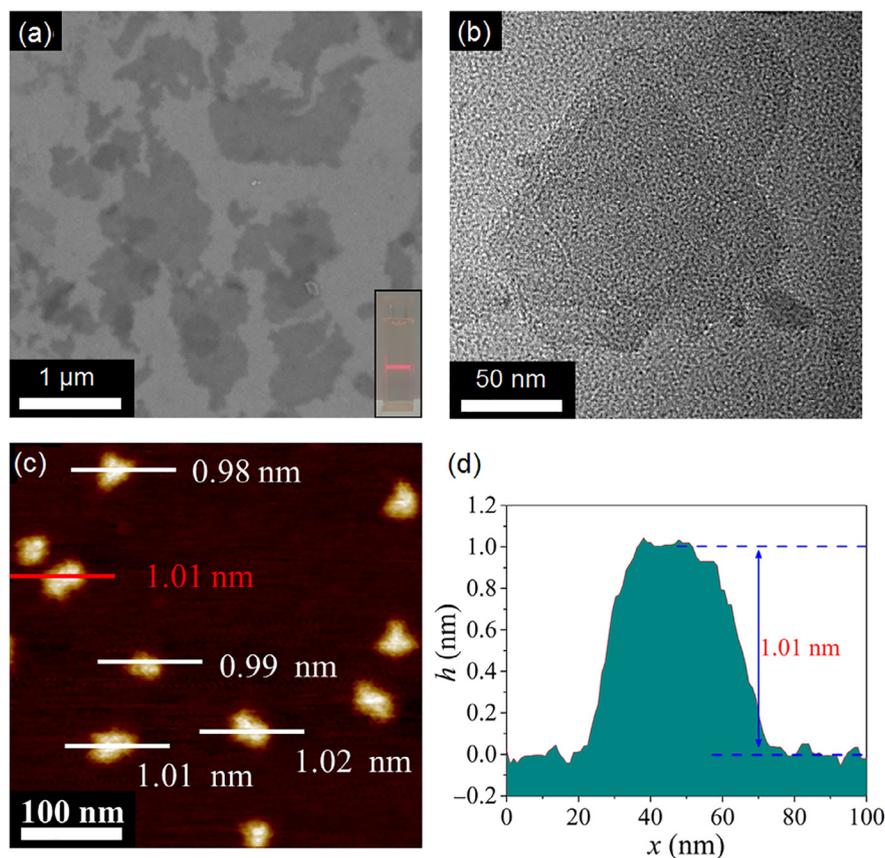


Fig. 3. (Color online) (a) SEM image (inset: an optical image showing the Tyndall effect), (b) TEM image, (c) AFM image with thickness measurements and (d) the corresponding height profile of NS.

Indeed, simple sonication of **4** in H₂O produced ultrathin nanosheets of [Cd₂(dnhbc)(H₂O)₄] (**NS**) with no PXRD peak (Fig. S14 online). Replacing **4** by **1** can also give the same result, confirming **4** as an intermediate of the evolution from **1** to **NS** (Fig. S14 online). IR spectra showed that, in **NS** the C–O(H) stretch vibration (1,248 cm⁻¹) and CO–H bending vibration (1,510 cm⁻¹) disappeared (Fig. S13 online), confirming the removal of the “pillars”. Elemental analysis gave a chemical composition consistent with the proposed formula [Cd₂(dnhbc)(H₂O)₄]. Scanning electron microscopy (SEM) demonstrated the thin slice morphology (Fig. 3a). Wrinkled or ruptured nanosheets were observed from transmission electron microscopy (TEM) (Fig. 3b). Atomic force microscopy (AFM) gave thickness of ca. 1.0 nm (Fig. 3c and d), corresponding well with the thickness of a single layer of [Cd₂(dnhbc)(H₂O)₄] containing coordination H₂O molecules, which can be simulated from **3** or **4** (Fig. S15 online). It should be noted that, the water dispersion of **NS** can be stable for at least 3 months (Fig. S16 online) [19].

In summary, we synthesized a new pillared-layer MOF which can be easily exfoliated to form ultrathin nanosheets in water. Interestingly, the exfoliation proceeds via multiple steps, and the intermediates can keep good crystallinity and even single-crystallinity, enabling us to directly visualize how water molecules attack the layers and replace the pillars. Similar exfoliation processes may occur in other systems when the layers have sufficiently high stability and the pillar-layer binding strengths are moderate. Preliminary study showed that the Mn analogue of **1** can display similar exfoliation processes.

Conflict of interest

The authors declare that they have no conflict of interest.

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Author contributions

Jie-Peng Zhang designed the research. Yun Li performed syntheses and most measurements. Jin Huang and Xue-Wen Zhang measured thermogravimetry and PXRD. Zong-Wen Mo measured IR spectra. Xiao-Ning Cheng measured SEM. Li Gong measured AFM. Yun Li, Dong-Dong Zhou and Jie-Peng Zhang analyzed data and wrote the manuscript.

Appendix A. Supplementary data

Supplementary data to this article can be found online at <https://doi.org/10.1016/j.scib.2019.05.011>.

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