



Original contribution

Non-destructive analysis of polymers and polymer-based materials by compact NMR

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ABSTRACT

Low-field nuclear magnetic resonance (NMR) based on permanent magnet technologies is currently experiencing a considerable growth of popularity in studying polymer materials. Various bulk properties can be probed with compact NMR tabletop instruments by placing the sample of interest inside the magnet. Contrary to this, compact NMR sensors with open geometries give access to depth-dependent properties of polymer samples and objects of different sizes and shapes truly non-destructively by performing measurements in the inhomogeneous stray-field outside the magnet system. Some of the sensors are also portable being thus well suited for onsite measurements. The gain of both bulk and depth-dependent microscopic properties are important for establishing improved structure-property relationships needed for the rational design of new polymer formulations. Selected recent applications will be presented to illustrate this potential of compact NMR.

1. Introduction

In the last decades, polymers revolutionized the progress of our society due their excellent properties tailored towards specific applications [1–5]. Such properties are achieved through ingenious formulations on which a polymer is mixed with various additives such plasticizers, stabilizers, and fillers. The interaction among the various ingredients as well as the structure, dynamics, and morphology of the building blocks at different time and length-scales critically impact the short- and long-term properties of the final product [6–9]. Furthermore, the effect of external factors during production and usage such as temperature, pressure, and deformation needs to be taken into account while designing a polymer product for a specific application. If these relationships would be fully understood, any macroscopic property could be predicted and controlled through the design of a defined microscopic structure, a dream of any chemist. Despite of much progress towards this goal in the last years through the combined efforts of scientists from different fields of activities, it still remains a major challenge.

In the last decades, huge efforts were done towards the development and the implementation of sophisticated analytical tools which can offer a detailed analysis of complex polymer structures both in liquid and solid-state in state of the art laboratories with the aim of establishing reliable structure-properties relationships [10–12]. Yet, there are also great efforts in developing low-cost compact table-top as well as portable analytical tools for onsite characterization of in-service

polymer products. Reasons behind are materials pre-screening, condition monitoring, real-time measurements to help make fast decisions. As an example, handheld portable sensors based on Fourier Transform Infrared Spectroscopy (FTIR) and RAMAN are already available to the market but their utilization in the field of non-destructive study of polymer materials is still rare [13–15].

Nuclear Magnetic Resonance is one of the analytical methods which substantially improved our understanding about the microscopic behavior of polymer materials and has largely contributed to current developments in polymer science [16–24]. While liquid-state NMR is the work horse of structural analysis of polymers in solution, its counterpart, solid-state NMR, is seen today as belonging to the top 10 newer analytical methods in polymer science [11]. Solid-state NMR measurements performed in a homogeneous and high-magnetic field delivers structural information with atomic resolution and enables characterizing the amplitude and the time scales of motions over broad ranges of length and time for both amorphous and semi-crystalline polymers [17–22]. The NMR techniques can overcome problems associated with various other analytical methods such as the need for long-range order as required by X-ray scattering. Moreover, depth- and lateral-dependent microscopic properties of a polymer product can be probed by NMR imaging technique while information about its interaction with liquid media is gained from diffusion studies [23–25]. As with other techniques, miniaturization also impacted the field of NMR [26]. As a consequence, table-top NMR sensors as well as portable NMR sensors with open and closed geometries and working at low magnetic

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fields are readily available today [26–28]. Such small and light compact sensors are an excellent alternative to the expensive and sophisticated high-field equipment for many applications as successfully demonstrated by investigations ranging from chemical science to cultural heritage [26–33]. Such sensors are able not only to measure relaxation, as demonstrated in earlier work, but also diffusion, images and high-resolution spectra thereby greatly expanding the area of applications [26–34].

This work reviews some recent applications of compact NMR in the field of polymer science and illustrates how NMR measurements with open and closed sensors of different magnetic field homogeneities provide simple solutions to challenging questions. Furthermore, the reliability of the method is discussed by comparing its results with those delivered by other classical analytical techniques.

2. Experimental methodology for polymer materials

Comprehensive reviews about compact low-field NMR instruments and measuring methods can be found, for example, in [27,28,34,35]. This minireview will highlight the instruments and experimental methods used in selected applications to polymer materials.

2.1. Compact NMR instruments

Fig. 1 depicts various compact low-field NMR sensors based on permanent magnet technologies which are currently used for the study of polymer materials. They include stray-field relaxometers with the Profile NMR-MOUSE being the most widely employed (Fig. 1a). With this sensor, the signal is acquired from a thin flat slice situated at a fixed distance above the surface of the magnet. It enables the measurement of a one-dimensional depth-profile by changing step by step the distance between the sensor and the object and with that the position of the sensitive slice inside the object. This is the best done automatically with the help of a high precision lift. A two-dimensional microscopic map of large polymer objects can be acquired by changing the lateral position of the sensitive slice. In combination with the depth-dependent information, a three-dimensional image can be obtained. Such sensors have an intrinsic strong static gradient somewhere between 10 T/m and 20 T/m enabling thus high-resolution depth-profiles and are in the most cases employed for relaxation and self-diffusion measurements. The

lateral resolution is generally lower than the depth resolution as it is determined by the size of the radio-frequency coil. Polymer solutions as well as solid amorphous and semi-crystalline polymer materials in form of powder, pellets, or more complicated shapes and much larger sizes such as car tires and pipelines can be measured truly non-destructively by placing them near by the NMR sensor [29,30,36–38].

Simple Halbach sensors (Fig. 1b) can also be used for the study of polymer materials. They are cylindrical magnets with the direction of the magnetic field transverse to the axis of the borehole and a very low stray-field magnetic field outside the borehole. The magnetic field is much homogeneous than for the stray-field sensors and the volume of the sensitive slice higher. They enable bulk relaxation measurements of samples which are either small enough or need to be cut in small pieces in order to fit inside the radio-frequency coil placed inside the magnet. Both the stray-field and the Halbach NMR sensors have a much lower magnetic field homogeneity than the desktop Bruker Minispec, another low-field NMR device with closed geometry and largely used for the study of polymer materials [19,21]. Yet, their much smaller size and weight compared to the Bruker Minispec make them well suited for measurements done under place restriction such as under the fume hood and for investigations outside the laboratory.

The highly homogeneous magnetic field of a SpinSolve spectrometer (Fig. 1c) is well suited for spectroscopy measurements of liquid samples being of primary use for the structural analysis of small molecules [34]. Small size and low installation price compared with the high-field NMR liquid-state devices compensate for their medium resolution spectra and enable also, for example, to be used as online detector for Size-Exclusion Chromatography (SEC-NMR) of polymers such as polymethylmethacrylate (PMMA) and polystyrene (PS) [39].

2.2. Measuring methods

The used measuring methods with the low-field sensors strongly depend on the homogeneity degree of the magnetic field. Relaxation measurements are most widely used with the sensors depicted in Fig. 1a and b. Information about the molecular dynamics of amorphous and semi-crystalline polymers is often gained with the help of multi-echo techniques such as CPMG (Carr, Purcell, Meiboom, Gill) pulse sequence which enables acquiring the whole relaxation decay within a single shot [40,41]. Instead of a CPMG, a multisolid-echo train employing the OW4

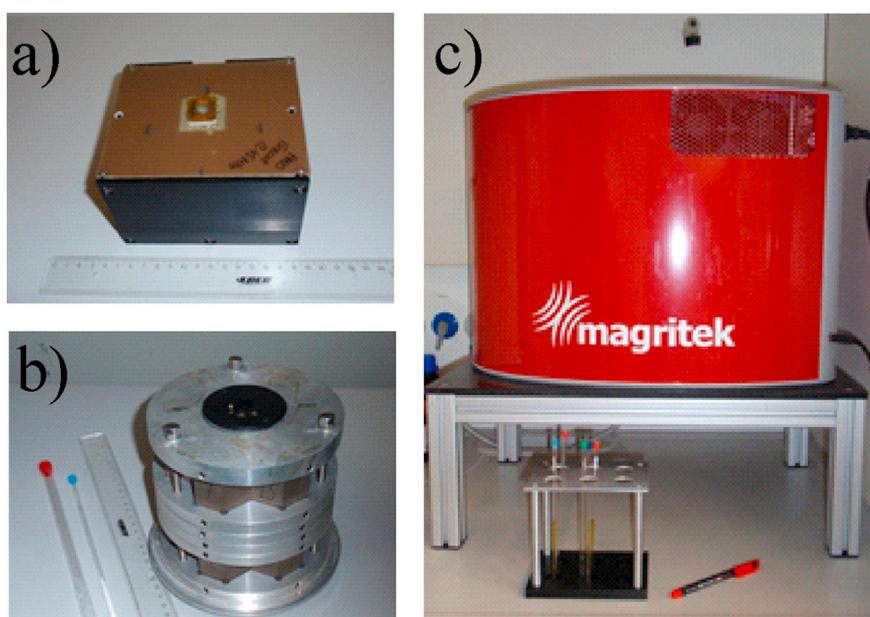


Fig. 1. Various compact low-field NMR sensors used for the study of polymer materials. (a) Profile NMR-MOUSE which enables non-destructive investigation of polymer samples and products with various shapes and size. (b) Halbach sensor. (c) SpinSolve spectrometer from Firma Magritek.

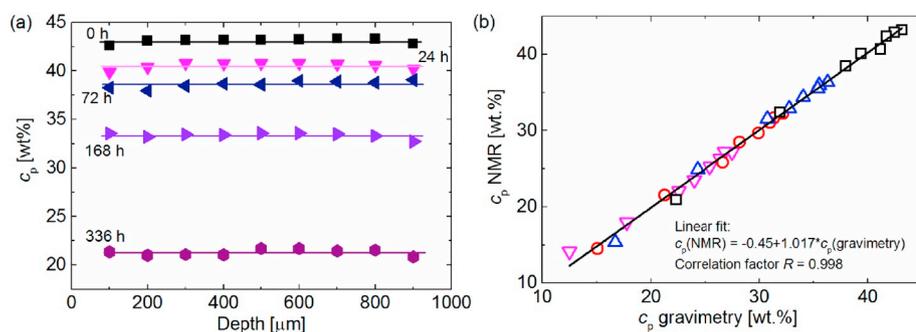


Fig. 2. (a) Local concentration c_p of ELATUR CH inside the PVC plates after different exposure times as quantified by NMR. The continuous lines serve as visual guide. (b) Correlation of the concentrations predicted by NMR with those directly determined by gravimetry. The NMR concentrations are the average of all concentrations across the plate thickness. The various symbols describe samples with different initial plasticizer concentrations and aged for various times under accelerated conditions. Adapted from [50].

(Ostroff-Waugh) pulse sequence is also sometimes employed [42]. The decay of the CPMG is characterized by an effective transverse relaxation time $T_{2\text{eff}}$ which is a mixture of the transverse relaxation time T_2 and the longitudinal relaxation time T_1 [34,35]. The decay of the OW4 is characterized also by an effective transverse relaxation time $T_{2\text{eff}}$ which in this case is a mixture of the transverse relaxation time T_2 and the longitudinal relaxation time in the rotating frame $T_{1\rho}$ [34,35].

The analysis of the experimental decays with analytical functions, usually exponential ones, delivers the values of the relaxation times $T_{2\text{eff}}$. An alternative procedure uses the inverse Laplace Transform to extract details about the distribution of the relaxation times [34]. The obtained values give information about the molecular motion. The higher the frequency and/or amplitude of the molecular motion is, the longer is the $T_{2\text{eff}}$. This dependency can then be used to quantify the molecular mobility of different regions in multiphase polymers given that the measurements are performed above the glass transition of the polymer under study. With this, also the proton fraction of each phase characterized by different mobility can be quantified giving thus information about the so called NMR phase composition. Identifying the most adequate NMR methodology and the best experimental conditions which provide the most accurate phase composition at low-field is still a field of intense research [43,44].

Both sensors enable also the quantification of proton longitudinal relaxation time T_1 usually with the help of the saturation recovery pulse sequence. The detection is in many cases done with an echo-train in order to improve the signal to noise. Also here an analysis using exponential functions or inverse Laplace Transform can be performed.

The strong and constant static field gradient of the Profile NMR-MOUSE allows as well probing the slow translational self-diffusion of small protonated penetrants in amorphous and semi-crystalline polymers [34,35,45,46]. For this, a constant gradient stimulated echo or a constant gradient Hahn-echo in conjunction with a CPMG echo-train for detection can be used [34,35,45,46]. The analysis of the diffusion curves can be done in terms of exponential functions or with the help of inverse Laplace Transform.

SpinSolve devices enable in most cases one-dimensional ^1H and ^{13}C spectra, 2D NMR measurements such as COSY and HETCOR and proton T_2 and T_1 relaxation measurements of samples in liquid-state [34]. One-dimensional ^1H NMR spectroscopy is often the method of choice for structural investigations due to its higher sensitivity. Such spectra are generally well suited for the structural analysis of small molecules and reaction monitoring [34]. Yet, their appearance increases in complexity for larger and more complicated molecules such as polymers and therefore one needs to make use of ^{13}C spectroscopy in the view of the larger chemical shift range. However, these measurements require much longer measuring times because of the low abundance of ^{13}C . For quality control purpose, a good alternative is the combination of proton spectra with chemometrics [47].

3. Applications

3.1. Quantification of local plasticizer concentration in PVC

Today, poly(vinyl chloride) (PVC) is frequently used in a large variety of applications including medical devices and packaging [48,49]. In most cases, PVC is used in a plasticized form with plasticizers making up to 80% of the mass. The content and type of plasticizer strongly control the macroscopic properties of the PVC product. As external plasticizers are used in most PVC products, their amount will diminish with increased usage time due to migration, a process which fails to be properly described by existing mathematical models. Furthermore, the loss of certain plasticizers raises questions about their impact on health and environment [48,49]. Moreover, there are studies pointing out the impact of the heterogeneous plasticizer distribution in a product on its properties, making thus the prediction of the long-term properties even more challenging. This complexity around the plasticized PVC products clearly demonstrates the need of analytical tools which can provide information about not only about the structure of the plasticizer but also about the bulk plasticizer content and its distribution.

In this context, a novel methodology using low-field NMR relaxometry was recently proposed for the non-destructive quantification of the local plasticizer concentration in PVC products within some couple of minutes [50]. The introduced approach makes use of correlation curves established between the transverse relaxation parameters measured with a profile NMR-MOUSE and the known bulk concentrations of plasticizer in fresh PVC plates with a homogeneous distribution of plasticizer. Slice-selective relaxation measurements at different depths inside a PVC sample with a known type of plasticizer enables the quantification of the local plasticizer content using the a priori established correlation curves, as demonstrated on samples with different aging times (Fig. 2a). The predicted concentrations from NMR show excellent agreement with those measured by gravimetry (Fig. 2b).

Furthermore, this study highlights the possibility of chemical identification of various plasticizers without the need of spectroscopic techniques using solely the above mentioned correlation curves [50]. Therefore, the proposed method has distinct advantages over other analytical tools currently applied in the study of plasticized PVC, including standard NMR, which either can probe only bulk or surface properties or need to extract the plasticizer for chemical identification and concentration quantification [51–53].

3.2. Solvents and polyethylene

The interaction between solvents and polymers plays a critical role in many applications including membrane technologies, pipeline transportation, packaging, coatings, and medical devices. Residual solvent can be trapped in the finished polymer products or the polymer products come into contact with solvents during many applications. The presence of the solvent can induce changes in the microscopic structure of the polymer including the molecular dynamics and morphology

through plasticization and solvent-induced crystallization and can cause an extraction of additives [6,8,54–56]. As a consequence, the macroscopic properties are altered. One the other hand, various polymer microstructure characteristics including crosslinking and morphology influence also the solvent uptake and evaporation [54–56]. This interplay is even more complex for semicrystalline polymers due to the presence of both crystalline and non-crystalline regions with the crystalline regions being normally not accessible to the solvent [57].

Low-field NMR provides fast and precise methodologies to online monitor the solvent ingress or its evaporation and to quantify the distribution of the solvent inside various polymer materials as well as the associated microscopic changes [58–63]. Furthermore, another great feature of NMR is its ability to directly measure the self-diffusion coefficients of the confined solvent inside the polymer materials [27,28,45,46]. This gives in turn a reliable way to identify microscopic properties controlling the diffusion behavior which is of key importance in improving the existing diffusion models [57].

In a recent study it was shown that low-field NMR relaxometry performed with a Halbach sensor (Fig. 1b) is an apt method to detect swelling-induced morphological changes in semicrystalline polymers with demonstration on polyethylene (PE) [64]. For this, the morphological changes of three commercial-brand PE grades, low-density polyethylene (LDPE), high-density PE (HDPE), and linear low-density PE (LLDPE), induced by swelling after different exposure times in n-hexane, chloroform, and xylene were investigated. All three solvents have been chosen due to their solubility parameters which are close to the solubility parameter of PE. It could be shown that the chain dynamics of the amorphous phase (Fig. 3a) decreases upon swelling and subsequent drying of the samples as a consequence of morphological arrangements leading to an increase in the degree of crystallinity. The extent of solvent-induced crystallization strongly depends on the type of solvent and PE grade and it is faster at higher temperatures. It seems to be determined by a complex interplay between the solubility parameters and the self-diffusion coefficients of the confined solvents. The low-field NMR observations could be confirmed by NMR measurements performed in a high-magnetic field and by Differential Scanning Calorimetry (DSC) results and are in agreement with literature findings on other polymer systems [65,66].

An improved understanding of the observed phenomena could also benefit from the existing fast and simple low-field NMR approach for the quantification of self-diffusion coefficients of small penetrants in semicrystalline polymers [46]. The extracted self-diffusion coefficients depend on the crystallinity of the samples and can also deliver information about the tortuosity of the samples [46].

The presence of oligomers in the final polymer product is reported to impact their thermo-mechanical properties [67,68]. Being preferentially localized in the non-crystalline regions of semicrystalline polymers, they can be easily extracted by the surrounding solvent and as a consequence their amount will change with increasing exposure time leading thereby to changes in various properties.

In this context, proton low-field NMR spectra acquired using

SpinSolve (Fig. 1c) has been for the first time employed to monitor and quantify the leaching of wax from LDPE pellets into n-hexane at room temperature [69]. Wax stands for the low-molecular weight fractions generated in any polymerization process. For the NMR study, the pellets were exposed to a defined amount of n-hexane. After different periods of time, the pellets were removed and dried at 70 °C until reaching the equilibrium in mass. Given that no solvent remains trapped in the dry samples, the change in their mass compared to the non-exposed pellets corresponds to the lost wax content. Parallel to this, ^1H spectroscopy in combination with the standard addition method was used to measure the wax content in remaining n-hexane/wax mixture. It was found out that the wax content increases with the swelling time and reaches the equilibrium after about 6 days of exposure (Fig. 3b). The NMR results show fairly good agreement with the results from gravimetry yet offering a much faster way of quantification.

3.3. Aging of semicrystalline polymers

During their lifetime, the properties of polymer products will degrade under the impact of various external factors such as temperature, UV-light, pressure, or solvents and this will eventually lead to their failure. It becomes thus clear that reliable predicting the safe working time is the key for their successful use in any application. While different technologies and solutions have been developed over the years to overcome limitations in lifetime, core challenges for specialists working today in the field of polymer aging are: the prediction of the lifetime, the understanding and identification of microscopic properties responsible for the failure of the polymer product, and the development and the implementation of portable analytical techniques for onsite condition monitoring [70–73].

Low-field NMR methods gained over the years in popularity for studying the aging of polymer materials due to its ability to characterize in detail the molecular network of both amorphous and semicrystalline polymers [74–80,82]. While structural changes can be only probed at high-field so far, various other microscopic details can also be gained using low-field NMR. In addition, sensors like the NMR-MOUSE (Fig. 1a) are of particular interest for onsite studies due to their portability and ability to perform truly non-destructive measurements.

The potential of single-sided NMR for condition monitoring of polymer materials has been, for example, evaluated by investigating the aging kinetics of silane crosslinked polyethylene (PEX) up to six months upon exposure to aggressive media used during oil and gas production and transportation [83]. To validate the NMR method, the same samples were investigated also by conventional condition monitoring tools including gravimetry, mechanical measurements, and DSC. All investigated properties show complicated changes with the aging time and exposure medium due to a complex morphological reorganization induced by a combination of effects including annealing, solvent-induced crystallization, and chain scissions and crosslinking. The mobility of the amorphous phase, as quantified with the help of the NMR relaxation measurements, shows the highest sensitivity to the aging

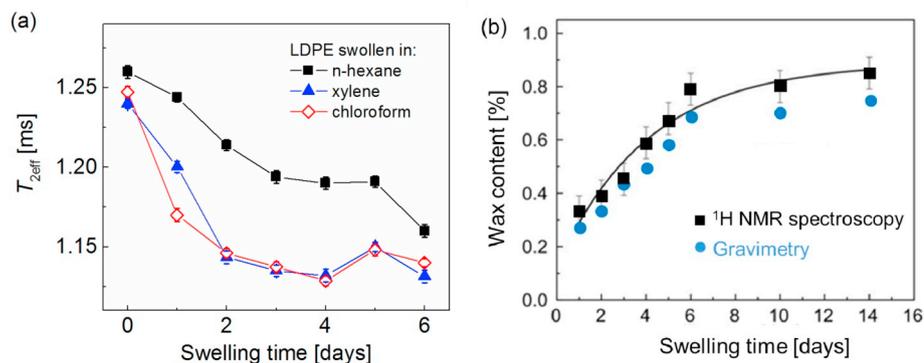


Fig. 3. (a) Changes in the proton relaxation times $T_{2\text{eff}}$ of the amorphous phase of LDPE pellets swelled for various times in three different solvents measured by low-field NMR relaxometry. Before being investigated by NMR, all samples were dried until no changes in the mass of the samples could be detected. The small differences in the relaxation times of the non-exposed samples can be attributed to small morphological differences among the studied pellets. Adapted from [64]. (b) Evaluation of the wax leaching from the same LDPE pellets by low-field ^1H spectroscopy. The wax content from gravimetry is shown for comparison purposes. Adapted from [69]. The lines serve in both figures as visual guide.

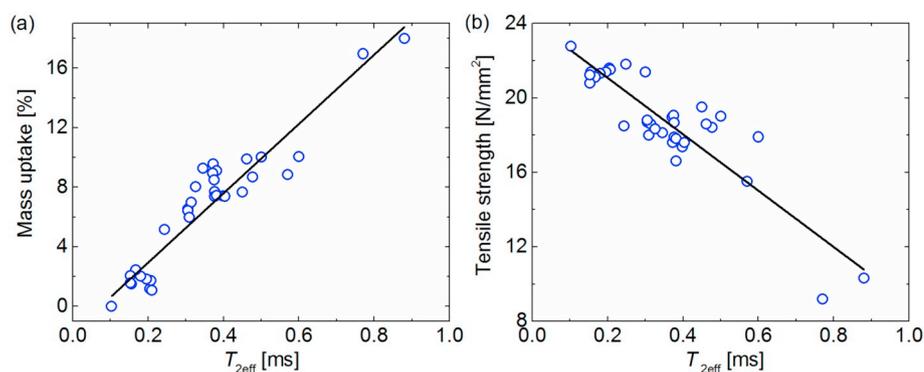


Fig. 4. Universal correlation curves of the proton relaxation time $T_{2\text{eff}}$ describing the mobility of the amorphous phase with the mass uptake (a) and tensile strength (b). The results obtained for all exposure times up to six months and all six exposure media are included. The correlation coefficient for the linear fits is higher than 0.8 despite of the fact that new pieces were used to generate the samples for each exposure condition.

conditions, in agreement with other aging studies [76,84]. Furthermore, it could be also demonstrated that the NMR results of all investigated samples correlate quasi linear with the results from mass uptake and mechanical measurements (Fig. 4), two classical destructive condition monitoring methods. This result is very important because of the current lack of analytical methods with can gain onsite information about the mechanical status of a polymer product and highlight the possibility of condition monitoring by simple non-destructive NMR relaxation measurements.

An improved understanding of the observed complex morphological reorganization in the presence of temperature and solvents is necessary to properly describe changes in the macroscopic properties. This would strongly benefit from disentangling which type of exposure is responsible for what change and how their effects combine. A comparison of the induced morphological changes in the same type of samples by either type of exposure is the way chosen in Ref. [69] using data primarily collected by NMR. The samples exposed to solvents were dried prior to the measurements in order to remove the effect of the solvent on the chain mobility of the amorphous phase as this microscopic parameter was also used for comparison in the view of its high sensitivity towards exposure conditions. Morphological changes including an increase in the fraction of the rigid domain at expense of the semi-rigid and mobile amorphous phases and a reduction in the chain mobility of the amorphous phase were detected for LDPE samples for both types of exposure. For thermal aging, these changes happen because of annealing and chemi-crystallization induced by chain scission. The observed morphological changes in the presence of solvents are of physical nature due to the solvent-induced crystallization as no chemical aging took place. Within the employed experimental conditions, the morphology changes induced by swelling and by elevated temperatures happen on different time scales with solvent-induced crystallization being much faster than temperature-induced crystallization.

Interpreting the changes in morphology of polymer products invoked by thermo-oxidation is further complicated by diffusion-limited oxidation (DLO) [71,85,86]. DLO induces depth-dependent morphological variations which in turn lead to depth-dependent variations of the macroscopic properties [71,85,86]. As an example, a thin oxidized layer, with a thickness of only few percent of the whole polymer thickness, significantly reduced the strain-at-break [86].

Non-destructive depth-dependent relaxation measurements with open NMR sensors such as the NMR-MOUSE gained in popularity for studying aging of polymer materials [87–89]. They offer an improve understanding of the local changes in the polymer network responsible for measured reduction in the mechanical properties [87–89].

For example, it was applied to investigate the depth-dependent changes during the thermo-oxidative aging at elevated temperatures of stabilized and non-stabilized polyamide 12 (PA12) plates [90]. It was found out that the morphological changes, which are due to a combination of annealing and chemical aging, are faster for the non-stabilized sample as for the stabilized one. They are the highest at surface and diminish than towards the middle of the plates. The interpretation of

the detected trends took advantage of the results from high-field NMR and could be confirmed by FTIR and viscosimetry. This highlights also the need of a multi-experimental approach to facilitate the understanding of the aging process.

4. Conclusions and outlook

Quantification of local plasticizer concentration in plasticized PVC, study of the interaction of semicrystalline polyethylene with solvents together with the characterization of the solvent-induced morphological changes, and aging induced bulk and depth-dependent changes of the molecular network are just few selected successful applications of compact NMR in polymer science. The excellent correlation of the reported NMR data with the results from classical analytical techniques including gravimetry and mechanical measurements is an additional confirmation of the great potential of the method to answer questions from fundamental and applied polymer research. On the other side, depending on the application, a combination of NMR with other analytical tools is highly desired for a deeper understanding of the observed microscopic and macroscopic behavior and for establishing structure-property relationships [87–91]. Furthermore, measurements with compact NMR would also strongly benefit from methods to increase the sensitivity. This could be achieved with the use of hyperpolarization methods such as DNP (Dynamic Nuclear Polarization) or parahydrogen [92–96], or with the help of multi-coil arrays [97].

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References

- [1] Andrady AL, Neal MA. Applications and societal benefits of plastics. *Philos T Roy Soc B* 2009;364:1977–84.
- [2] Geise GM, Lee HS, Miller DJ, Freeman BD, Mcgrath JE, Paul DR. Water purification by membranes: the role of polymer science. *J Polym Sci B Polym Phys* 2010;48: (1685–18).
- [3] Halliwell S. *Polymers in building and construction*. iSmithers Rapra Publishing; 2002.
- [4] Marsh K, Bugusu B. Food packaging—roles, materials, and environmental issues. *J Food Sci* 2007;72:R39.
- [5] Galizia M, Chi WS, Smith ZP, Merkel TC, Baker RW, Freeman BD. Polymers and mixed matrix membranes for gas and vapor separation: a review and prospective opportunities. *Macromolecules* 2017;50:7809–43.
- [6] Van Krevelen DW. *Properties of polymers*. 4th ed. Elsevier B.V.; 2009. (Revised by Te Nijenhuis K).
- [7] Meijer HEH, Govaert LE. Mechanical performance of polymer systems: the relation between structure and properties. *Prog Polym Sci* 2005;30:915–38.
- [8] Geise GM, Paul DR, Freeman BD. Fundamental water and salt transport properties of polymeric materials. *Prog Polym Sci* 2014;39:1–42.
- [9] Chodák I. Properties of crosslinked polyolefin-based materials. *Prog Polym Sci* 1995;20:1165–99.
- [10] Liu Y, Wang Z, Zhang X. Characterization of supramolecular polymers. *Chem Soc Rev* 2012;41:5922–32.

- [11] Lodge TP. *Macromolecules* 2017;50:9525–7.
- [12] Yang R. *Analytical methods for polymer characterization*. Boca Raton: CRC Press 2018.
- [13] Capitán-Vallveya LF, Palma AJ. Recent developments in handheld and portable optosensing—a review. *Anal Chim Acta* 2011;696:27–46.
- [14] <https://www.agilent.com/cs/library/applications/5991-6976EN.pdf>, Accessed date: 20 August 2018.
- [15] Cucci C, Bartolozzi G, Marchiafava V, Piccolo M, Richardson E. Study of semi-synthetic plastic objects of historic interest using non-invasive total reflectance FT-IR. *Microchem J* 2016;124:889–97.
- [16] NMR spectroscopy of polymers: Innovative strategies for complex macromolecules. In: Cheng HN, Asakura T, English A, editors. ACS Symposium Series. 1077. 2011.
- [17] Spiess HW. 50th anniversary perspective: the importance of NMR spectroscopy to macromolecular science. *Macromolecules* 2017;50:1761–77.
- [18] Hansen MR, Graf R, Spiess HW. Interplay of structure and dynamics in functional macromolecular and supramolecular systems as revealed by magnetic resonance spectroscopy. *Chem Rev* 2016;116:1272–308.
- [19] Saalwächter K. Proton multiple-quantum NMR for the study of chain dynamics and structural constraints in polymeric soft materials. *Prog Nucl Magn Reson Spect* 2007;51:1–35.
- [20] Schmidt-Rohr K, Spiess HW. *Multidimensional solid-state NMR and polymers*. London: Academic Press; 1994.
- [21] Litvinov V. Molecular mobility and phase composition in polyolefins: from fundamental to applied research. *NMR spectroscopy of polymers: innovative strategies for complex macromolecules*. 1077. 2011. p. 179–90.
- [22] Asano A. NMR relaxation studies of elastomers. *Annu Rep NMR Spectrosc* 2015;86:1–72.
- [23] Blümich B. *NMR imaging of materials*. Oxford: Clarendon Press; 2000.
- [24] Price WS. *NMR studies of translational motion*. Cambridge: Cambridge University Press; 2009.
- [25] Walderhaug H, Söderman O, Topgaard D. Self-diffusion in polymer systems studied by magnetic field-gradient spin-echo NMR methods. *Prog Nucl Magn Reson Spect* 2010;56:406–25.
- [26] Zalesskiy SS, Danieli E, Blümich B, Ananikov VP. Miniaturization of NMR systems: desktop spectrometers, microcoil spectroscopy, and “NMR on a chip” for chemistry, biochemistry, and industry. *Chem Rev* 2014;114:5641–94.
- [27] Blümich B, Perlo J, Casanova F. Mobile single-sided NMR. *Prog Nucl Magn Reson Spect* 2008;52:197–269.
- [28] Mitchell J, Gladden L, Chandrasekera T, Fordham E. Low-field permanent magnets for industrial process and quality control. *Prog Nucl Magn Reson Spect* 2014;76:1–60.
- [29] Blümich B, Singh K. Desktop NMR and its applications from materials science to organic chemistry. *Angew Chem Int Ed* 2018;57:6996–7010.
- [30] Rehorn C, Blümich B. Cultural heritage studies with mobile NMR. *Angew Chem Int Ed* 2018;57:7304–12.
- [31] Oligschläger D, Rehorn C, Lehmkuhl S, Adams M, Adams A, Blümich B. A size-adjustable radiofrequency coil for investigating plants in a Halbach magnet. *J Magn Reson* 2017;278:80–7.
- [32] Chen JJ, Kong X, Sumida K, Manumpil MA, Long JR, Reimer JA. Ex situ NMR relaxometry of metal-organic frameworks for rapid surface-area screening. *Angew Chem Int Ed* 2013;52:12043–6.
- [33] Meyer K, Kern S, Zientek N, Guthausen G, Maiwald M. Process control with compact NMR. *TrAC* 2016;83:39–52.
- [34] Blümich B. Introduction to compact NMR: a review of methods. *TrAC* 2016;83:2–11.
- [35] Adams A. Analysis of solid technical polymers by compact NMR. *TrAC* 2016;83:107–19.
- [36] Adams A, Blümich B. Single-sided NMR of semicrystalline polymers. *Macromol Symp* 2013;327:29–38.
- [37] Blümich B, Casanova F, Buda A, Kremer K, Wegener T. Mobile NMR for analysis of polyethylene pipes. *Acta Phys Pol A* 2005;108:13.
- [38] Adams A, Adams M, Blümich B, Rooks JH, Hilgert O, Zimmermann S. Nondestructive testing procedure for evaluation of fracture-mechanically relevant abnormalities in partially crystalline polymers. *3R Int* 2010;4:216–25.
- [39] Höpfner J, Ratzsch KF, Botha C, Wilhelm M. Medium resolution ¹H-NMR at 62 MHz as a new chemically sensitive online detector for size-exclusion chromatography (SEC-NMR). *Macromol Rapid Commun* 2018;39:1700766–73.
- [40] Carr HY, Purcell EM. Effects of diffusion on free precession in nuclear magnetic resonance experiments. *Phys Rev* 1954;94:630–8.
- [41] Meiboom S, Gill D. Modified spin-echo method for measuring nuclear relaxation times. *Rev Sci Instrum* 1958;29:688–91.
- [42] Ostroff E, Waugh J. Multiple spin echoes and spin locking in solids. *Phys Rev Lett* 1996;16:1097–9.
- [43] Maus A, Hertlein C, Saalwächter K. A robust proton NMR method to investigate hard/soft ratios, crystallinity, and component mobility in polymers. *Macromol Chem Phys* 2006;207:1150–8.
- [44] Rätzsch V, Haas M, Özen MB, Ratzsch KF, Riazi K, Kauffmann-Weiss S, et al. Polymer crystallinity and crystallization kinetics via benchtop ¹H NMR relaxometry: revisited method, data analysis, and experiments on common polymers. *Polymer* 2018;145:162–73.
- [45] Rata D, Casanova F, Perlo J, Demco D, Blümich B. Self-diffusion measurements by a mobile single-sided NMR sensor with improved magnetic field gradient. *J Magn Reson* 2006;180:229–35.
- [46] Kwamen R, Blümich B, Adams A. Estimation of self-diffusion coefficients of small penetrants in semicrystalline polymers using single-sided NMR. *Macromol Rapid Commun* 2012;33:943–7.
- [47] Singh K, Blümich B. Compact low-field NMR spectroscopy and chemometrics: a tool box for quality control of raw rubber. *Polymer* 2018;141:154–65.
- [48] Rahman M, Brazel CS. The plasticizer market: an assessment of traditional plasticizers and research trends to meet new challenges. *Prog Polym Sci* 2004;29:1223–48.
- [49] Latini G, Ferri M, Chiellini F. Materials degradation in PVC medical devices, DEHP leaching and neonatal outcomes. *Curr Med Chem* 2010;17:2979–89.
- [50] Adams A, Kwamen R, Woldt B, Graß M. Nondestructive quantification of local plasticizer concentration in PVC by ¹H NMR relaxometry. *Macromol Rapid Commun* 2015;36:2171–5.
- [51] Bernard L, Décaudin B, Lecoer M, Richard D, Bourdeaux D, Cuffe R, et al. Analytical methods for the determination of DEHP plasticizer alternatives present in medical devices: a review. *Talanta* 2014;129:39–54.
- [52] Sommer S, Koch M, Adams A. Terahertz time-domain spectroscopy of plasticized poly(vinylchloride). *Anal Chem* 2018;90:2409–13.
- [53] Barendswaard W, Litvinov V, Souren F, Scherrenberg R, Gondard C, Colemonts C. Crystallinity and microstructure of plasticized poly(vinylchloride). A ¹³C and ¹H solid state NMR study. *Macromolecules* 1999;32:167–80.
- [54] Rozanski A, Galeski A. Plastic yielding of semicrystalline polymers affected by amorphous phase. *Int J Plast* 2013;41:14–29.
- [55] Smith GD, Karlsson K, Gedde UW. Modeling of antioxidant loss from polyolefins in hot-water applications. I: model and application to medium density polyethylene pipes. *Polym Eng Sci* 1992;32:658–67.
- [56] Gedde UW, Viebke J, Leijström H, Ifwarson M. Long-term properties of hot-water polyolefin pipes—a review. *Polym Eng Sci* 1994;34:1773–87.
- [57] Hedenqvist M, Gedde UW. Diffusion of small-molecule penetrants in semicrystalline polymers. *Prog Polym Sci* 1996;21:299–333.
- [58] Velasco MI, Silletta EV, Gomez CG, Strumia MC, Stapf S, Monti GA, et al. Spatially resolved monitoring of drying of hierarchical porous organic networks. *Langmuir* 2016;32:2067–74.
- [59] Ghoshal S, Denner P, Stapf S, Mattea C. Study of the formation of poly(vinylalcohol) films. *Macromolecules* 2012;45:1913–23.
- [60] Nestle N, Quero F, Tissier N. Drying and film formation's first steps in polymer dispersions – insights from a parallel gravimetric experiment and profile NMR study. *Microporous Mesoporous Mater* 2015;205:79–82.
- [61] Reuvers N, Huinink H, Fischer H, Adan O. Quantitative water uptake study in thin nylon-6 films with NMR imaging. *Macromolecules* 2012;45:1937–45.
- [62] Reuvers NJW, Huinink HP, Adan OCG. Plasticization lags behind water migration in nylon-6: an NMR imaging and relaxation study. *Polymer* 2015;63:127–33.
- [63] Chassé W, Schlögl S, Riess G, Saalwächter K. Inhomogeneities and local chain stretching in partially swollen networks. *Soft Matter* 2013;9:6943–54.
- [64] Teymouri Y, Adams A, Blümich B. Compact low-field NMR: unmasking morphological changes from solvent-induced crystallization in polyethylene. *Eur Polym J* 2016;80:48–57.
- [65] Vittoria V, Riva F. Solvent-induced crystallization of quenched isotactic polypropylene in different liquids. *Macromolecules* 1986;19:1975–9.
- [66] Ouyang H, Lee WH, Ouyang W, Shiu ST, Wu TM. Solvent-induced crystallization in poly(ethylene terephthalate) during mass transport: mechanism and boundary condition. *Macromolecules* 2004;37:7719–23.
- [67] Rozanski A, Idczak R. Influence of non-polymeric substances localized in the amorphous phase on selected properties of semicrystalline polymers. *Eur Polym J* 2015;69:186–200.
- [68] Djokovic V, Mshali TN, Luyt AS. The influence of wax content on the physical properties of low-density polyethylene-wax blends. *Polym Int* 2003;52:999–1004.
- [69] Teymouri Y, Adams A, Blümich B. Impact of exposure conditions on the morphology of polyethylene by compact NMR. *Macromolecular Symposia* 2018;378:1600156.
- [70] Celina M, Gillen KT, Assink RA. Accelerated aging and lifetime prediction: review of non-Arrhenius behaviour due to two competing processes. *Polym Degrad Stab* 2005;90:395–404.
- [71] Celina MC. Review of polymer oxidation and its relationship with materials performance and lifetime prediction. *Polym Degrad Stab* 2013;98:2419–29.
- [72] Fayolle B, Colin X, Audouin L, Verdu J. Mechanism of degradation induced embrittlement in polyethylene. *Polym Degrad Stab* 2007;92:231–8.
- [73] Laycock B, Nikolic M, Colwell JM, Gauthier E, Halley P, Bottle S, et al. Lifetime prediction of biodegradable polymers. *Prog Polym Sci* 2017;71:144–89.
- [74] Da Silva PSRC, Tavares MIB. The use of relaxometry to evaluate the aging process in hybrid HIPS nanocomposites. *Polym Test* 2015;48:115–9.
- [75] Chinn S, DeTeresa S, Sawvel A, Shields A, Balazs B, Maxwell RS. Chemical origins of permanent set in a peroxide cured filled silicone elastomer-tensile and ¹H NMR analysis. *Polym Degrad Stab* 2006;91:555–64.
- [76] Litvinov V, Soliman M. The effect of storage of poly(propylene) pipes under hydrostatic pressure and elevated temperatures on the morphology, molecular mobility and failure behaviour. *Polymer* 2005;46:3077–89.
- [77] Paul J, Hansen EW, Roots J. Probing the molecular dynamics in XLPE aged at different temperatures by ¹H NMR relaxation time measurements. *Polym Degrad Stab* 2012;97:2403–11.
- [78] Badea E, Sendrea C, Carşote C, Adams A, Blümich B, Iovu H. Unilateral NMR and thermal microscopy studies of vegetable tanned leather exposed to dehydrothermal treatment and light irradiation. *Microchem J* 2016;129:158–65.
- [79] Sendrea C, Badea E, Adams A. Unilateral NMR and micro DSC study of artificially aged parchments. *Rev Roum Chim* 2017;68:1780–5.
- [80] Shi X, Wang J, Stapf S, Mattea C, Li W, Yang Y. Effects of thermo-oxidative aging on chain mobility, phase composition, and mechanical behavior of high density polyethylene. *Polym Eng Sci* 2011;51:2171–7.
- [82] Teymouri Y, Kwamen R, Blümich B. Aging and degradation of LDPE by compact

- NMR. *Macromol Mater Eng* 2015;300:1063–70.
- [83] Adams A, Piechatzek A, Schmitt G, Siegmund G. Single-sided nuclear magnetic resonance for condition monitoring of cross-linked polyethylene exposed to aggressive media. *Anal Chim Acta* 2015;887:163–71.
- [84] Sun N, Wenzel M, Adams A. Morphology of high-density polyethylene pipes stored under hydrostatic pressure at elevated temperature. *Polymer* 2014;55:3792–800.
- [85] Gijssman P, Dong W, Quintana A, Celina M. Influence of temperature and stabilization on oxygen diffusion limited oxidation profiles of polyamide 6. *Polym Degrad Stabil* 2016;130:83–96.
- [86] Wei XF, Kallio KJ, Bruder S, Bellander M, Kausch HH, Gedde UW, et al. Diffusion-limited oxidation of polyamide: three stages of fracture behavior. *Polym Degrad Stabil* 2018;154:73–83.
- [87] Pourmand P, Hedenqvist MS, Furó I, Gedde UW. Deterioration of highly filled EPDM rubber by thermal ageing in air: kinetics and non-destructive monitoring. *Polym Test* 2017;64:267–76.
- [88] Pourmand P, Linde E, Hedenqvist MS, Furo I, Dvinskikh SV, Gedde UW. Profiling of thermally aged EPDM seals using portable NMR, indenter measurements and IR spectroscopy facilitating separation of different deterioration mechanisms. *Polym Test* 2016;53:77–84.
- [89] Kehlet C, Catalano A, Dittmer J. Degradation of natural rubber in works of art studied by unilateral NMR and high field NMR spectroscopy. *Polym Degrad Stabil* 2014;107:270–6.
- [90] Zhang J, Adams A. Understanding thermal aging of non-stabilized and stabilized polyamide 12 using ^1H solid-state NMR. *Polym Degrad Stabil* 2016;134:169–78.
- [91] Röntzsch V, Wilhelm M, Guthausen G. Hyphenated low-field NMR techniques: combining NMR with NIR, GPC/SEC and rheometry. *Magn Reson Chem* 2016;54:494–501.
- [92] Ouari O, Phan T, Ziarelli F, Casano G, Aussenac F, Thureau P, et al. Improved structural elucidation of synthetic polymers by dynamic nuclear polarization solid-state NMR spectroscopy. *ACS Macro Lett* 2013;2:715–9.
- [93] Gizatullin B, Neudert O, Stapf S, Mattea C. Dynamic nuclear polarization fast field cycling method for the selective study of molecular dynamics in block copolymers. *Chemphyschem* 2017;18:2347–56.
- [94] Neudert O, Reh M, Spiess HW, Münnemann K. X-band DNP hyperpolarization of viscous liquids and polymer melts. *Macromol Rapid Comm* 2015;36:885–9.
- [95] Halse ME. Perspectives for hyperpolarisation in compact NMR. *TrAc* 2016;83:76–83.
- [96] Richardson PM, Jackson S, Parrott AJ, Nordon A, Duckett SB, Halse ME. A simple hand-held magnet array for efficient and reproducible SABRE hyperpolarisation using manual sample shaking. *Magn Reson Chem* 2018;56:641–50.
- [97] Oligschläger D, Lehmkuhl S, Watzlaw J, Benders S, de Boever E, Rehorn C, et al. Miniaturized multi-coil arrays for functional planar imaging with a single-sided NMR sensor. *J Magn Reson* 2015;254:10–8.